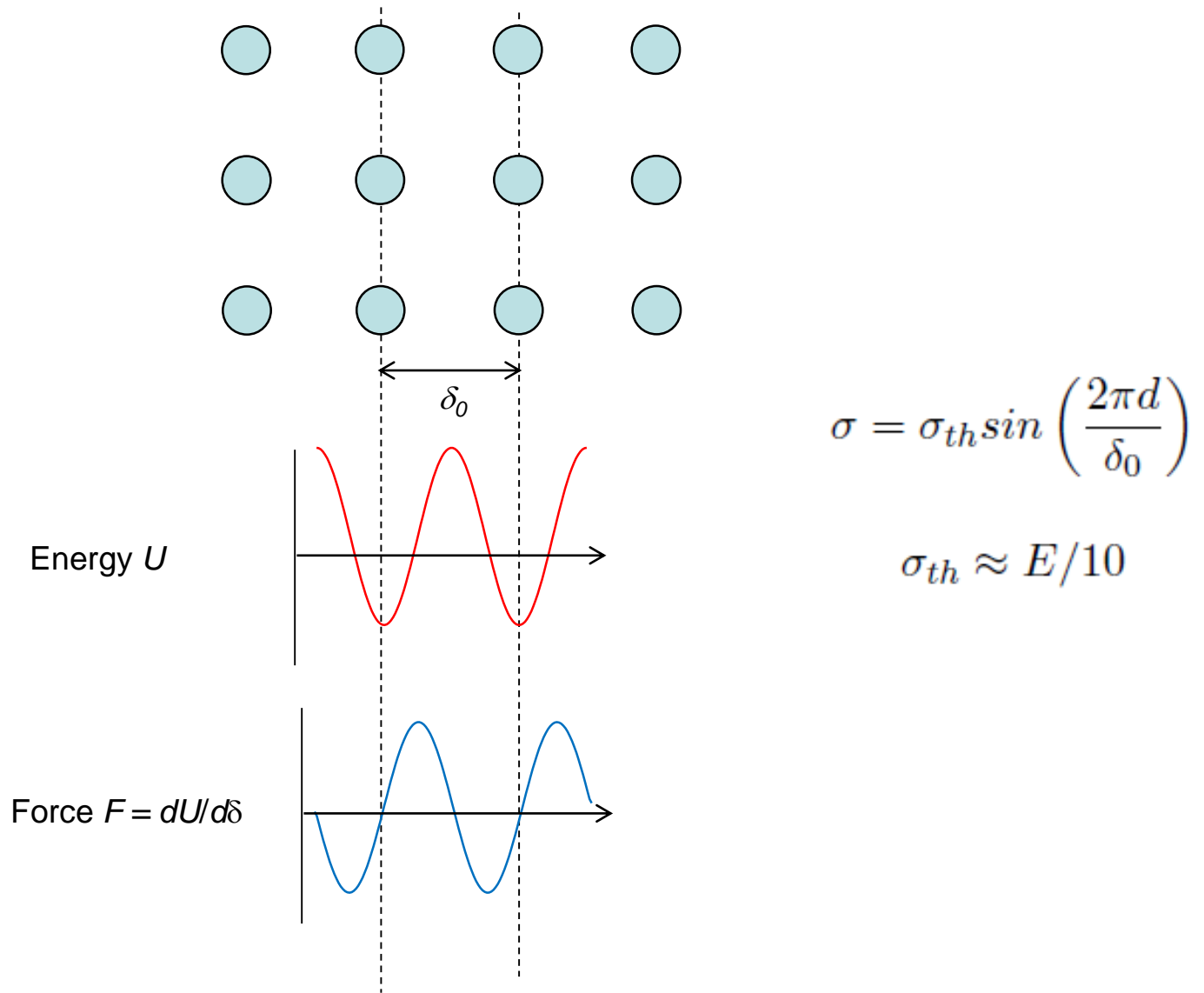


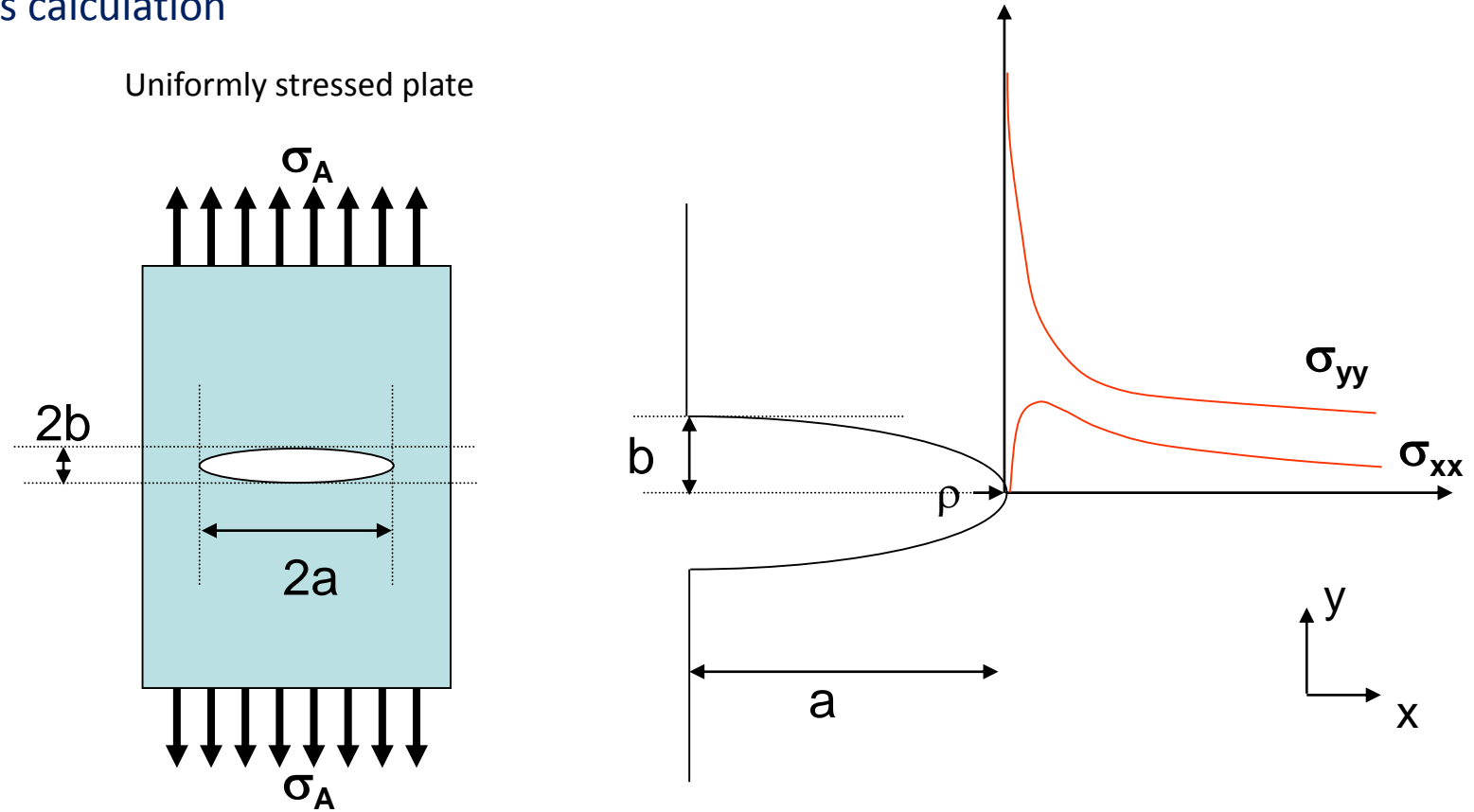
INTRODUCTION TO LINEAR ELASTIC FRACTURE MECHANICS

Theoretical strength of a perfectly ordered crystalline lattice



Stress concentration at the vicinity of an elliptical hole

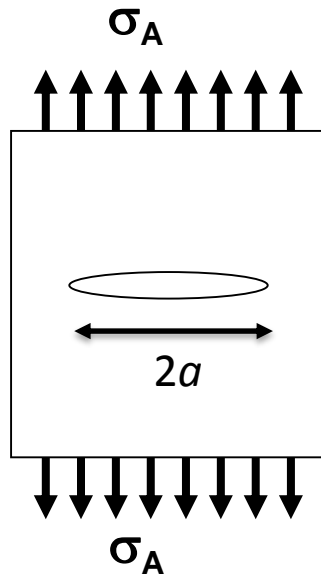
- Inglis calculation



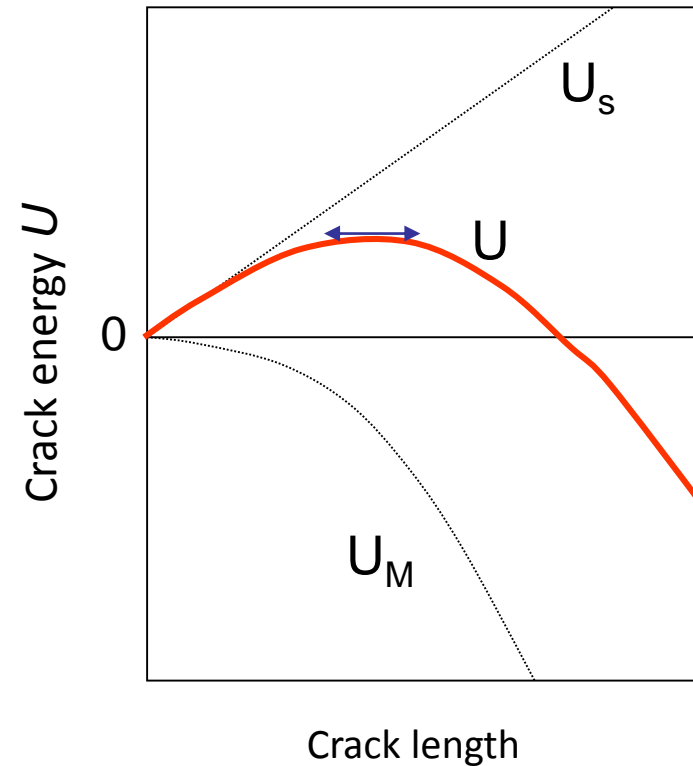
$$\rho = \frac{b^2}{c}$$

$$\frac{\sigma_c}{\sigma_a} \approx 2 \frac{c}{b} = 2 \sqrt{\frac{c}{\rho}}$$

Crack in uniform tension



Infinitely narrow elliptical cavity
Remote uniform tensile stress field σ_A

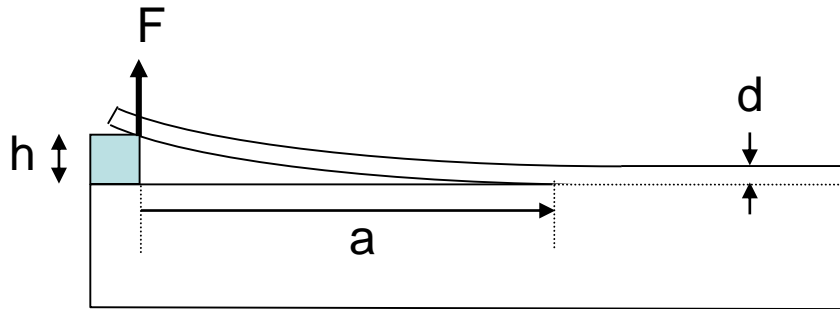


$$U(a) = -\pi a^2 \sigma_A^2 / E' + 4a\gamma$$

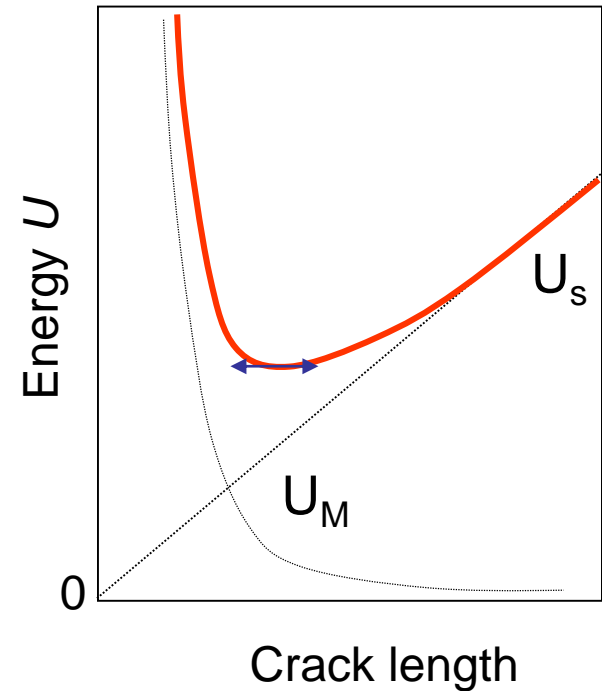
Unstable configuration

Cleavage by a wedge

- Obreimoff's experiment (cleavage of mica)

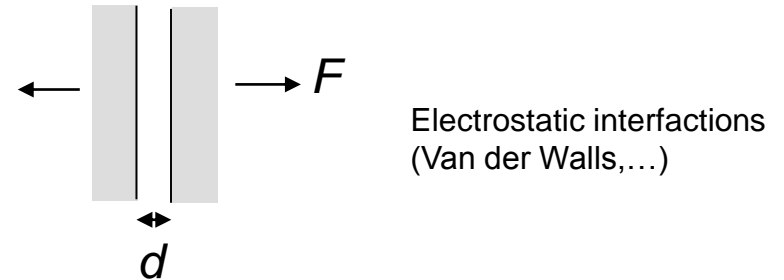
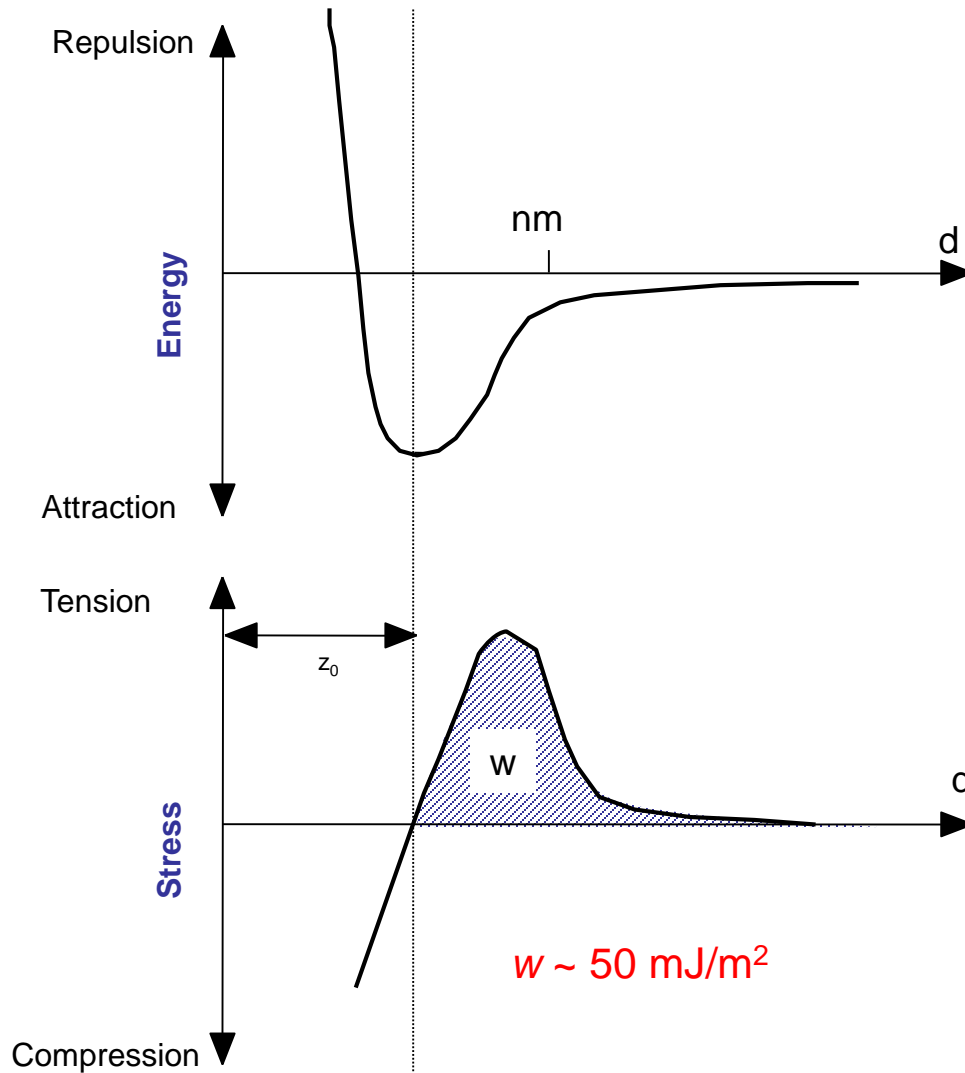


$$U(a) = \frac{Ed^3h^2}{8c^3} + 2a\gamma$$



Stable configuration

Stress-separation function for two atom planes



The Dupré work of adhesion:

$$W = \gamma_1 + \gamma_2 - \gamma_{12}$$

Homogeneous solid $w = 2\gamma$

Equilibrium conditions !
No chemistry !

Condition for equilibrium fracture : the Griffith criterion

- Energy balance for crack extension

$$dU = dU_M + dU_S = -GdA + wdA$$

where $G = -\left(\frac{\partial U_E}{\partial A} + \frac{\partial U_P}{\partial A}\right)_P = \left(\frac{\partial U_E}{\partial A}\right)_\delta$ Strain energy release rate

$$\frac{\partial U}{\partial A} = 0$$

$$G = w$$

EQUILIBRIUM condition !!

No dissipative processes !!

In most practical situations $G_c \gg w$

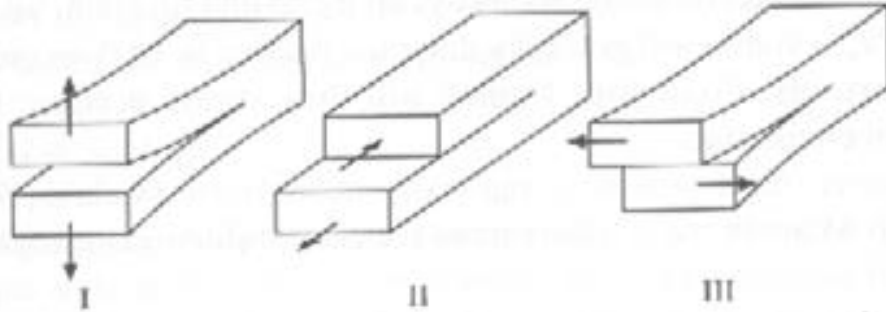


- Energy dissipation at the crack tip (plasticity, viscoelasticity) and/or in the bulk
- Dependence of actual fracture micro-mechanisms at the crack tip (process zone)

.....

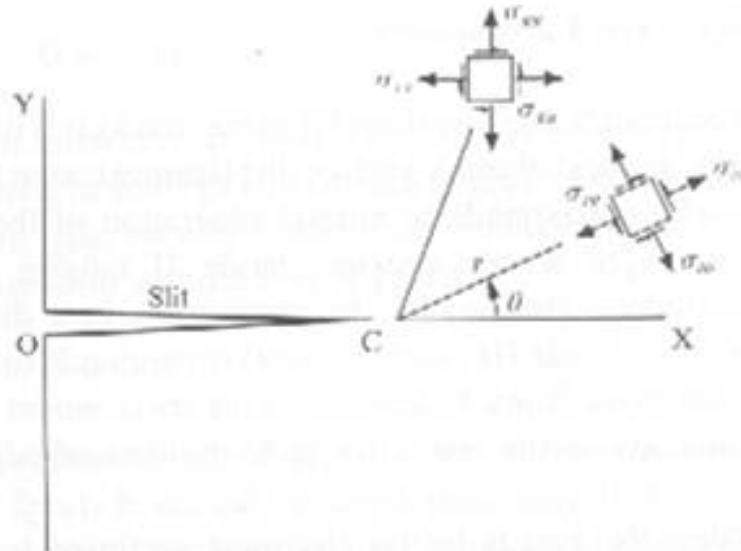
Irwin's approach : Linear elastic crack-tip fields

- The three modes of fracture



- I : opening mode
- II: sliding mode (in plane shear)
- III: tearing mode (out of plane shear)

- Irwin slit-crack tip in rectangular and polar coordinates



Stress-intensity factors K

Mode I crack loading

$$\sigma_{ij} = \frac{K}{\sqrt{2\pi r}} f_{ij}(\theta)$$

$$u_i = \frac{K}{2E} \sqrt{\frac{r}{2\pi}} f_i(\theta)$$

$$K = \chi \sigma \sqrt{a}$$

$K=f$ (geometry, applied loading)

$$\begin{aligned} \kappa &= (3-\nu)/(1+\nu), & \nu' &= 0, & \nu'' &= \nu, & \text{(plane stress)} \\ \kappa &= (3-4\nu), & \nu' &= \nu, & \nu'' &= 0, & \text{(plane strain).} \end{aligned}$$

Mode I:

$$\begin{Bmatrix} \sigma_{xx} \\ \sigma_{yy} \\ \sigma_{xy} \end{Bmatrix} = \frac{K_I}{(2\pi r)^{1/2}} \begin{Bmatrix} \cos(\theta/2)[1 - \sin(\theta/2)\sin(3\theta/2)] \\ \cos(\theta/2)[1 + \sin(\theta/2)\sin(3\theta/2)] \\ \sin(\theta/2)\cos(\theta/2)\cos(3\theta/2) \end{Bmatrix}$$

$$\begin{Bmatrix} \sigma_{rr} \\ \sigma_{\theta\theta} \\ \sigma_{r\theta} \end{Bmatrix} = \frac{K_I}{(2\pi r)^{1/2}} \begin{Bmatrix} \cos(\theta/2)[1 + \sin^2(\theta/2)] \\ \cos^2(\theta/2) \\ \sin(\theta/2)\cos^2(\theta/2) \end{Bmatrix}$$

$$\sigma_{xx} - \nu'(\sigma_{xx} + \sigma_{yy}) = \nu'(\sigma_{rr} + \sigma_{\theta\theta})$$

$$\sigma_{xx} - \sigma_{yy} = \sigma_{rr} - \sigma_{\theta\theta} = 0$$

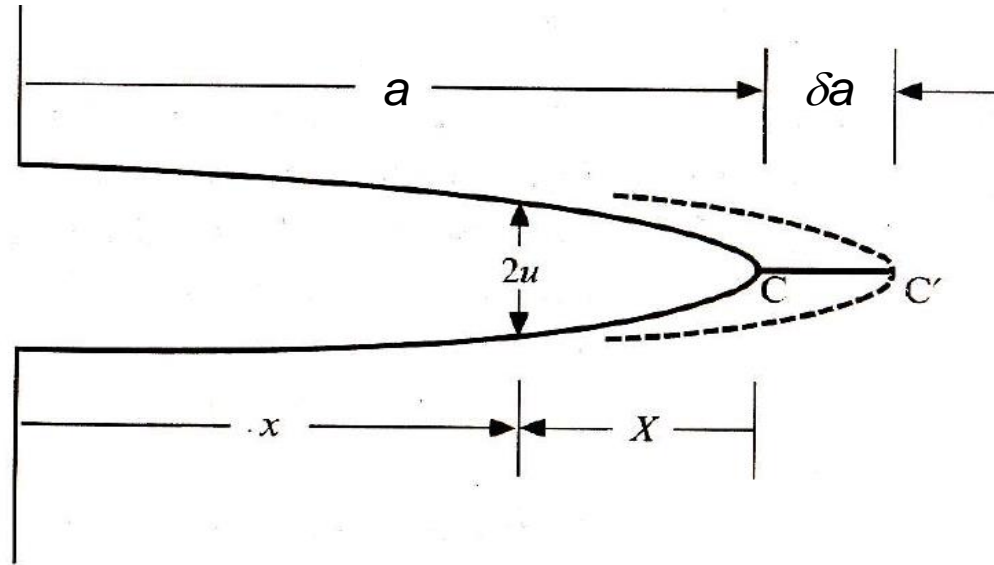
$$\begin{Bmatrix} u_r \\ u_\theta \end{Bmatrix} = \frac{K_I}{2E} \left[\frac{r}{2\pi} \right]^{1/2} \begin{Bmatrix} (1+\nu)[(2\kappa-1)\cos(\theta/2) - \cos(3\theta/2)] \\ (1+\nu)[(2\kappa+1)\sin(\theta/2) - \sin(3\theta/2)] \end{Bmatrix}$$

$$\begin{Bmatrix} u_r \\ u_\theta \end{Bmatrix} = \frac{K_I}{2E} \left[\frac{r}{2\pi} \right]^{1/2} \begin{Bmatrix} (1+\nu)[(2\kappa-1)\cos(\theta/2) - \cos(3\theta/2)] \\ (1+\nu)[- (2\kappa+1)\sin(\theta/2) + \sin(3\theta/2)] \end{Bmatrix}$$

$$u_z = -(\nu'' z/E)(\sigma_{xx} + \sigma_{yy}) = -(\nu'' z/E)(\sigma_{rr} + \sigma_{\theta\theta}).$$

Equivalence of G and K parameters

- Extension and closure of the crack increment CC'



Fixed grip ($u=\text{const}$) calculation of the strain-energy release:

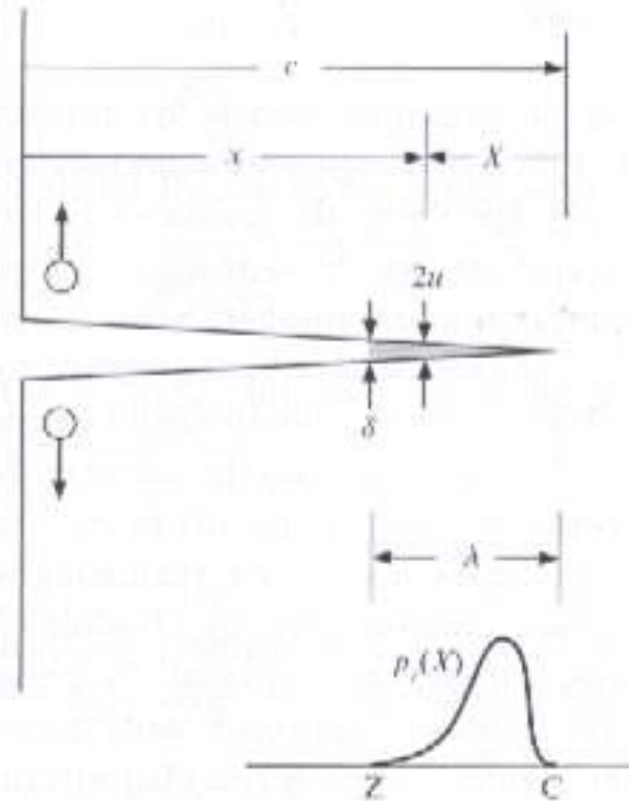
$$\delta U_E = 2 \int_{a+\delta a}^a \frac{1}{2} (\sigma_{yy} u_y + \sigma_{xy} u_x + \sigma_{zy} u_z) dx$$

$$G = - \left(\frac{\partial U_E}{\partial c} \right)_u = K_I^2 / E' + K_{II}^2 / E' + K_{III}^2 (1 + \nu) / E$$

plane strain : $E' = E / (1 - \nu^2)$; plane stress : $E' = E$

Cohesive zone models

Dugdale, Barenblatt



$$\lambda \ll c \ll L$$

- Non linear cohesive stress $p(X,u)$ acting accross the walls of the slip
- Superposition of the K fields arising from the 'external' and 'internal contributions'

Physical origins of the cohesive zone

- Surface forces acting across the crack faces
- Plastic deformation at the crack tip (Dugdale model)
- Crack bridging in crazes (polymer fracture)
-

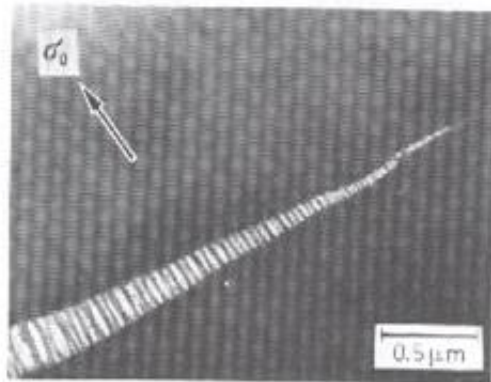


Fig. 9.9. Newly formed craze in thin slice cut from uncrazed bulk polystyrene; craze growing from left to right in a direction perpendicular to that of the uniaxial tensile stress (Courtesy D. Hull [106]).

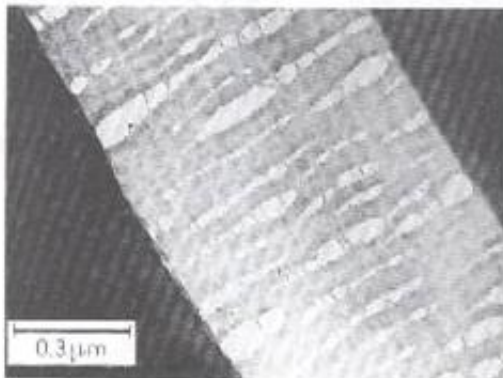


Fig. 9.10. Electron micrograph of the central section of a craze grown as that in Figure 9.9 (Courtesy D. Hull [106]).

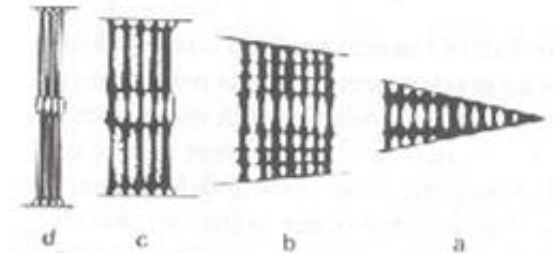
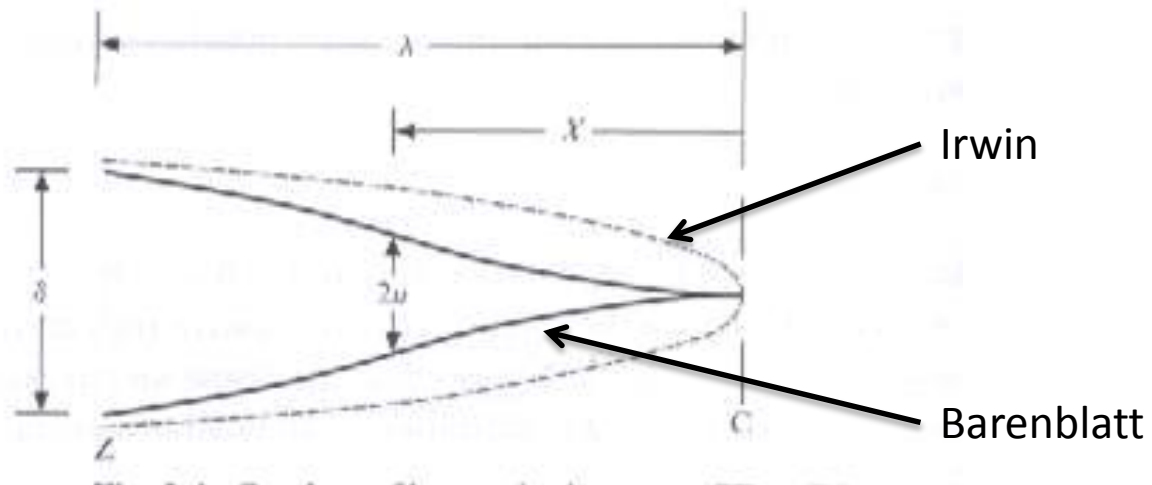


Fig. 9.11. Schematic diagram of the variation in polystyrene craze structure with increasing craze width; the angle at the tip of the craze is exaggerated and the scale in region *d* is larger than that of *a* to *c*. (Courtesy D. Hull [115]).

Barenblatt model: crack-opening profile



Hyp: $p(X, u)_{X \rightarrow c} \rightarrow 0$

$$u(X)_{X \rightarrow c} \propto X^{3/2}$$

$$\epsilon_{X \rightarrow c} = du/dX \rightarrow 0$$

Irwin

$$u(X) \propto X^{1/2}$$

$$\epsilon_{X \rightarrow c} \rightarrow \infty$$

Path independent integrals about crack tip : The J integral approach

Eshelby, Rice

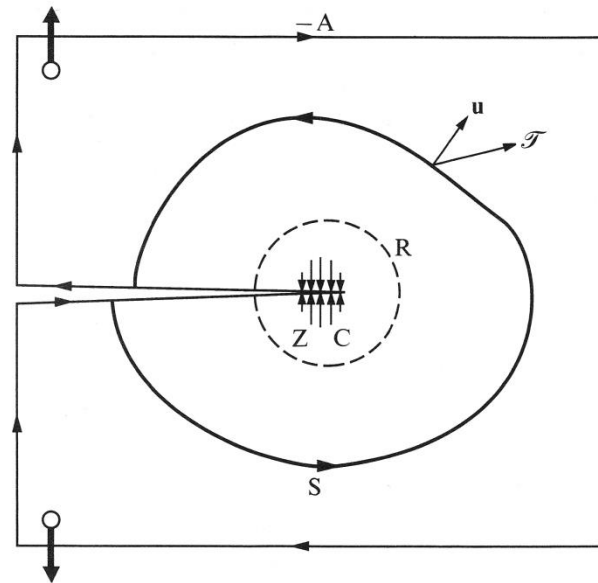


Fig. 3.6. Line integration path S about crack tip C in plane static system, unit thickness. J -integral around closed circuit S-A (arrows) is zero. Special J -integrals are taken around polar circuit R and cohesion zone CZ.

$$U_M = U_E + U_P = \int_A \Phi dA - \int_S \chi \cdot \mathbf{u} ds$$

$$-dU_M/da = \int_S [\Phi dy - \chi \cdot (\partial \mathbf{u} / \partial x) ds] = J$$

J integral → holds for any *reversible* deformation response *linear* or non *linear*
 → path independent

Brittle materials : Hertzian fracture / I

Sphere on flat contacts

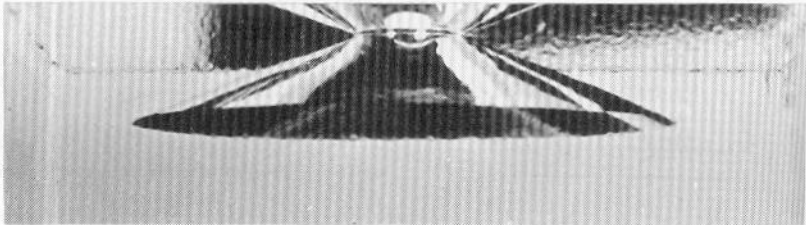


Fig. 8.3. Cone crack in soda-lime glass. Photographed under load ($P = 40$ kN) from cylindrical punch, optical micrograph (block edge length 50 mm). Crack makes angle $\approx 22^\circ$ to free surface. (After Roesler, F. C. (1956) *Proc. Phys. Soc. Lond.* **B69** 981.)

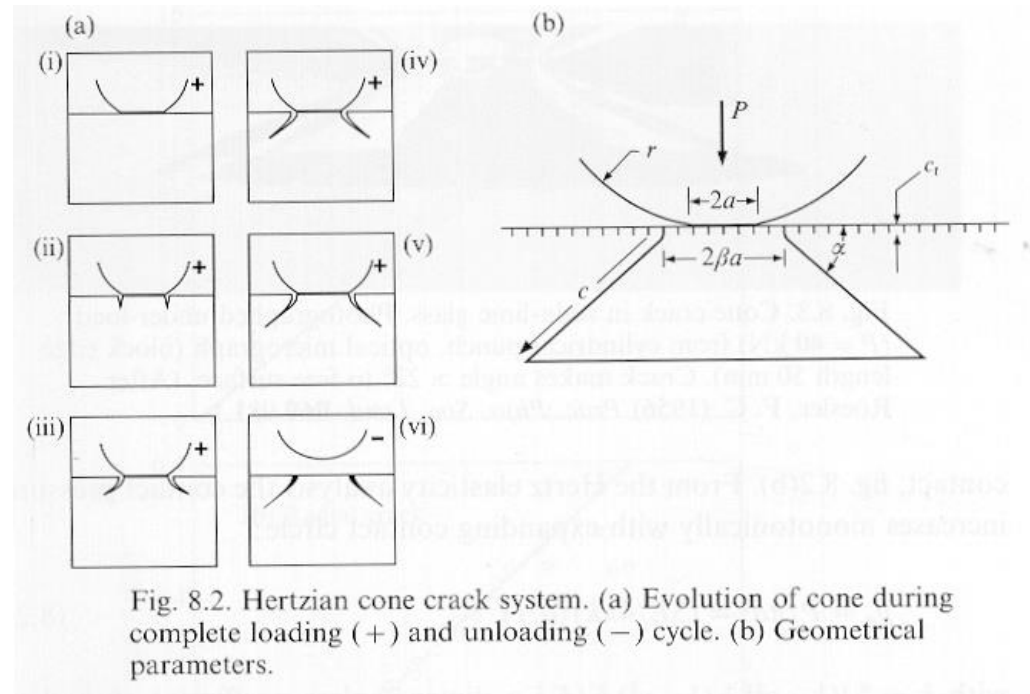
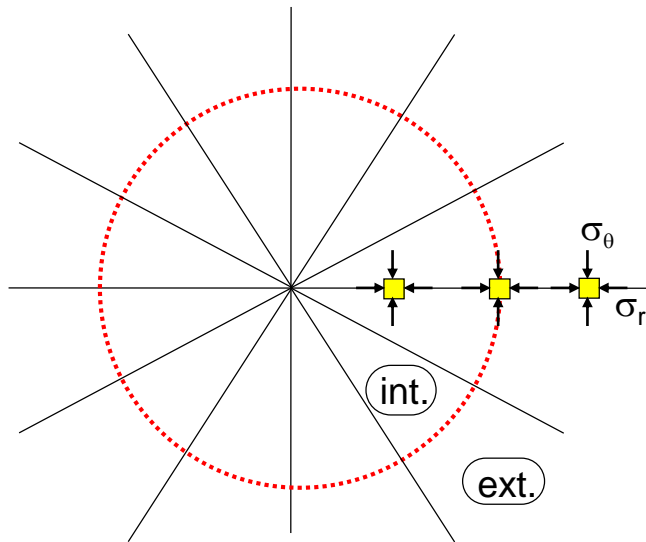


Fig. 8.2. Hertzian cone crack system. (a) Evolution of cone during complete loading (+) and unloading (-) cycle. (b) Geometrical parameters.

- (ii) Crack initiation at the edge of the contact
- (iii) Stable crack propagation
- (iv) Unstable crack propagation : formation of the Hertzian cone
- (v) Stable propagation of the cone
- (vi) Crack closing during unloading

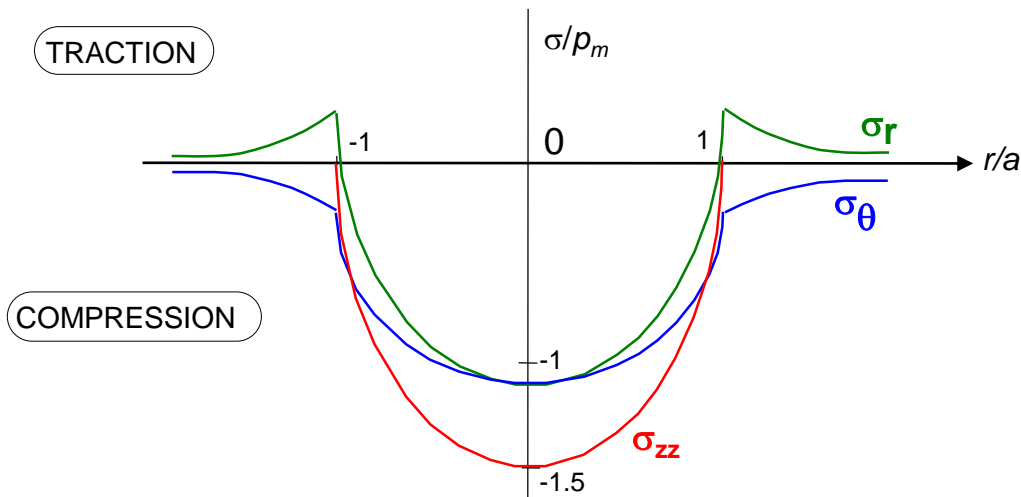
Hertz contact : surface tractions



Maximum tensile stress:

$$\sigma_R = \frac{1}{2}(1-2\nu)p_m \approx 0.1p_m$$

p_m = mean contact pressure



Point surface loading: sub-surface stress fields

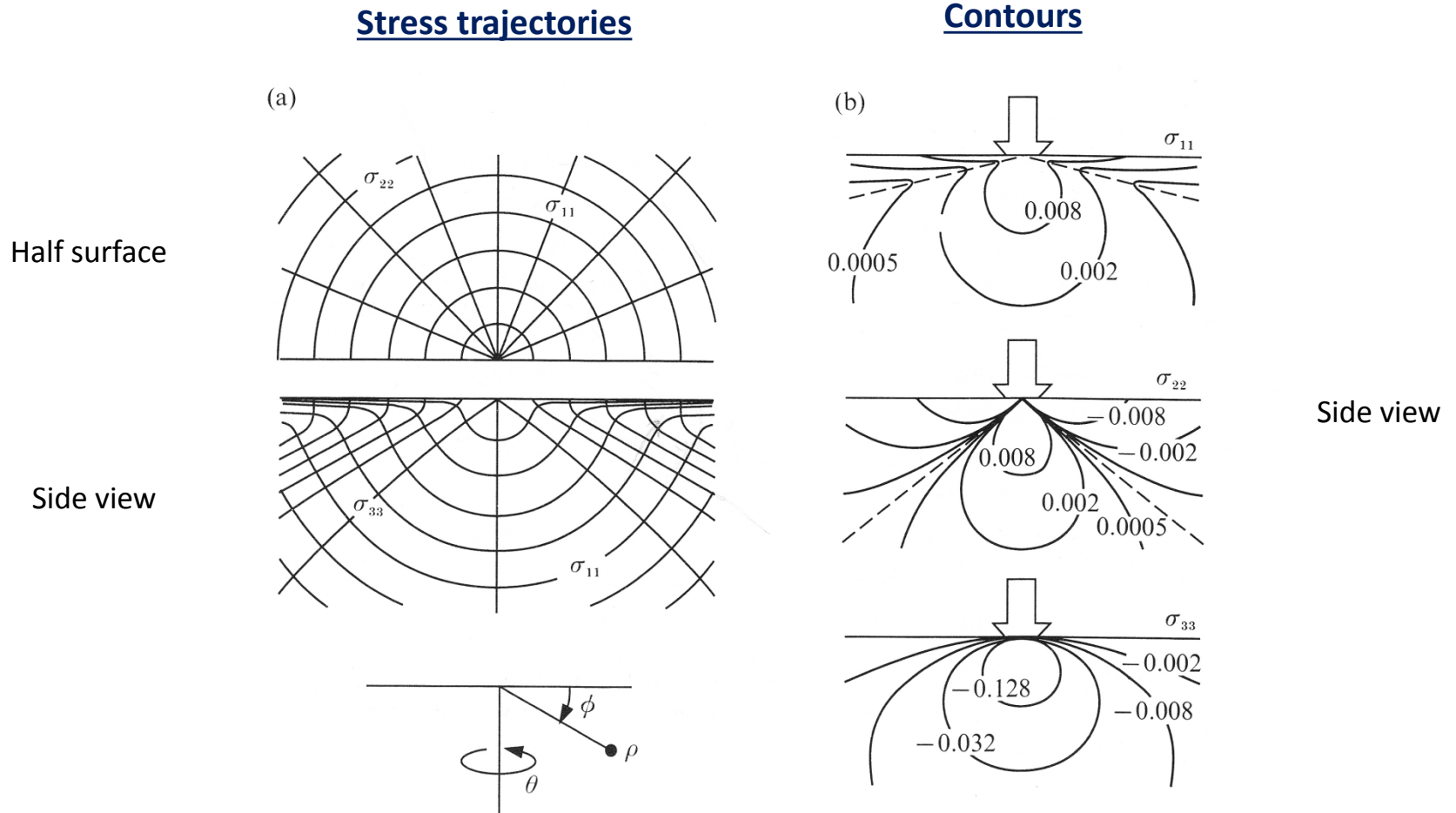


Fig. 8.1. Boussinesq field, for principal normal stresses σ_{11} , σ_{22} and σ_{33} . (a) Stress trajectories, half-surface view (top) and side view (bottom). (b) Contours, side view. Unit of contact stress is p_0 , contact diameter $2a$ (arrow). Note sharp minimum in $\sigma_{11}(\phi)$ and zero in $\sigma_{22}(\phi)$, dashed lines. Plotted for $\nu = 0.25$. (See Johnson, K. L. (1985) *Contact Mechanics*. Cambridge University Press, Cambridge, Ch. 3.)

Critical load for the propagation of the Hertzian cone crack

• Stress-based approach: $\sigma_R = \frac{1}{2}(1-2\nu)p_m \rightarrow P_c \propto R^2$

Experimentally $P_c \propto R!$

- Linear elastic fracture mechanics analysis approach

Non homogeneous stress field: the stress intensity factor is evolving as along the crack path

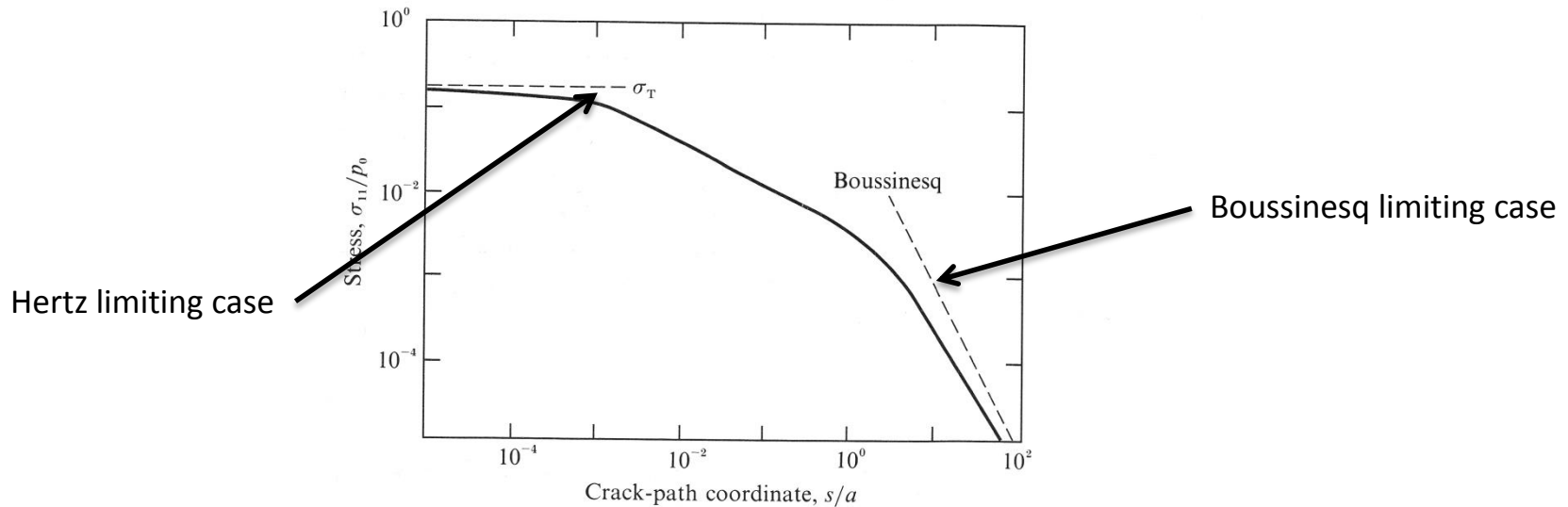
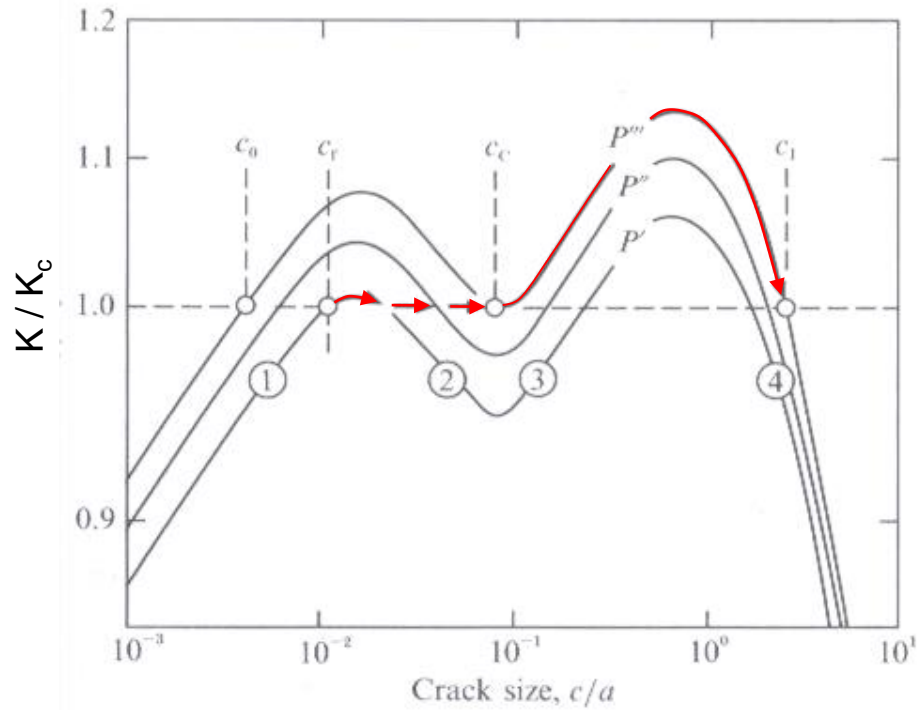


Fig. 8.30. Reduced plot of principal tensile stress σ_{11} vs distance downward along $\sigma_{22}-\sigma_{33}$ stress trajectory surface in Hertzian field. Asymptotes (dashed lines) correspond to bounds of uniform field $\sigma_{11} = \sigma_T$ (Griffith flaw, $c \ll a$) and Boussinesq inverse-square field (Roesler cone, $c \gg a$). Note rapid stress falloff below surface, $\sigma_{11}/\sigma_T < 0.1$ at $s/a = 0.1$. Plots for $\beta = 1$, $\nu = \frac{1}{3}$. (After Frank, F. C. & Lawn, B. R. (1967) *Proc. Roy. Soc. Lond.* **A299** 291; Lawn, B. R. & Wilshaw, T. R. (1975) *J. Mater. Sci.* **10** 1049.)

Stability criterion for a Hertzian crack



Branchs(1) et (3) : instable propagation
 Branchs (2) et (4) : stable propagation

Crack of length c_c :Hertzian cone formation

Critical load

$$P_c \propto R \frac{K_c^2}{E^*}$$

Brittle materials : cone indentation

Vickers

- Plastic deformation
- Radial and median cracks

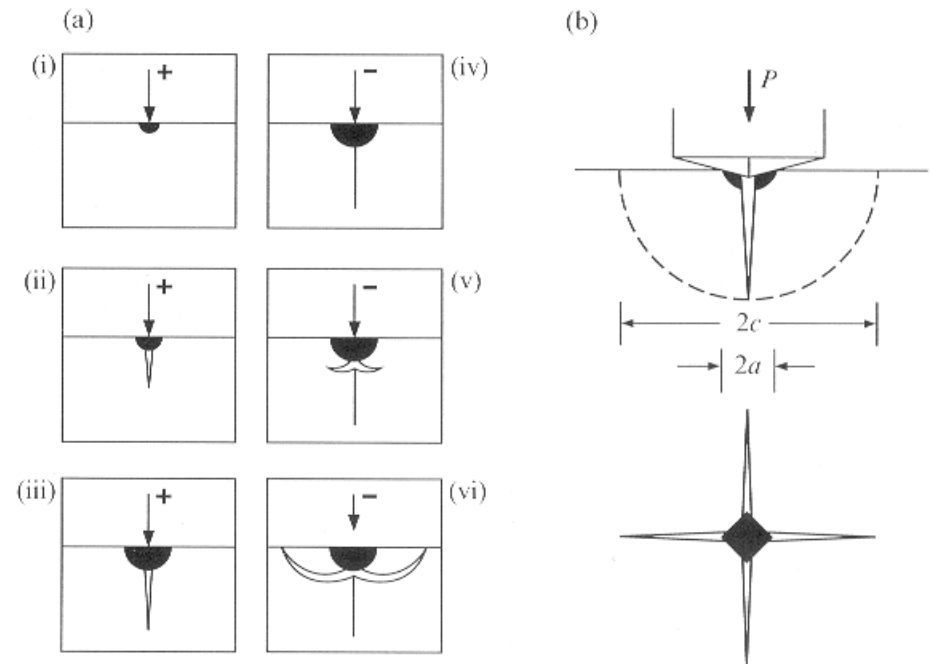
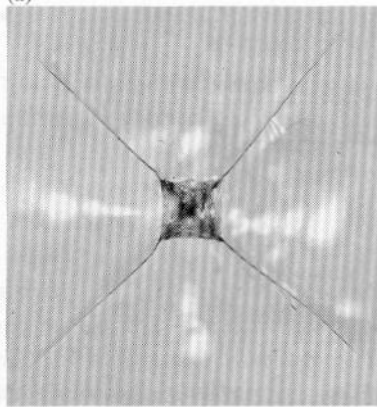


Fig. 8.7. Radial–median and lateral crack systems. (a) Evolution during complete loading (+) and unloading (–) cycle. Dark region denotes irreversible deformation zone. (b) Geometrical parameters of radial system.

Residual stresses during unloading:

→ Nucleation and propagation of median cracks

→ Surface propagation of radial cracks

Toughness measurements using indentation

$$\sigma_R Y \sqrt{a_c} = K_c$$

- Sphère / plan $c \gg a$ $P > P_c$

$$K_c = \chi \frac{P}{c^{3/2}}$$

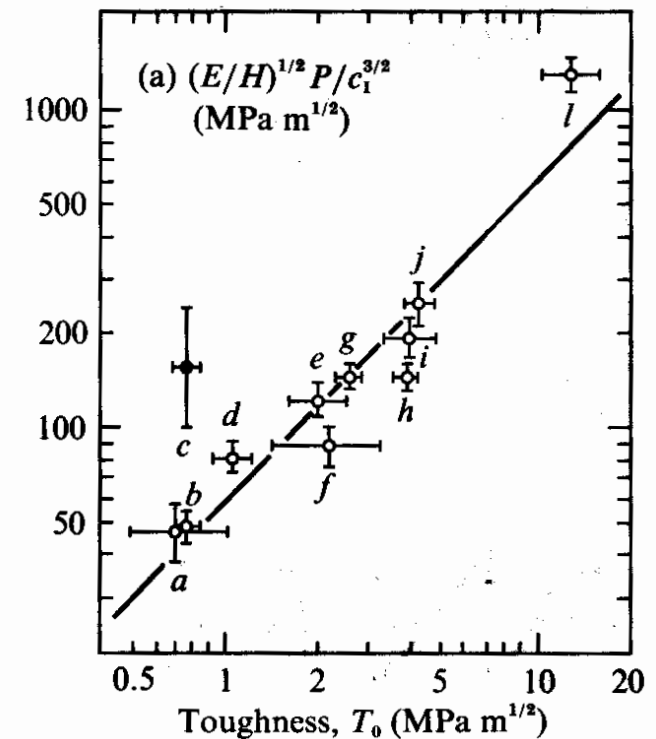
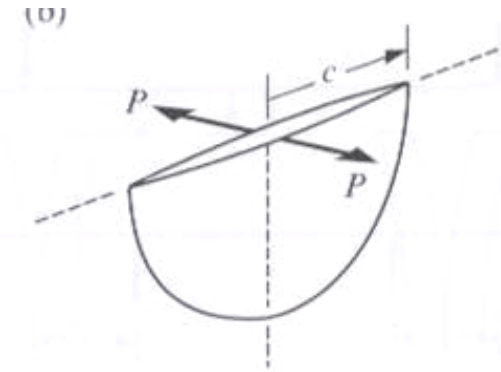
χ : non dimensional parameter dependent on Poisson's ratio

- Vickers

$$K_c = \xi \left(\frac{E}{H} \right)^{1/2} \frac{P}{c^{3/2}}$$

$$\xi = \xi_0 (\cot \Phi)^{2/3}$$

ξ_0 : non dimensional constant function of the strain distribution



Vickers indentation

Determination of fracture toughness from indentation testing

Half-penny crack $\sigma_R \sqrt{\pi a_c} = K_c$

- sphere on flat $c \gg a$

$$K_c = \chi \frac{P}{c^{3/2}}$$

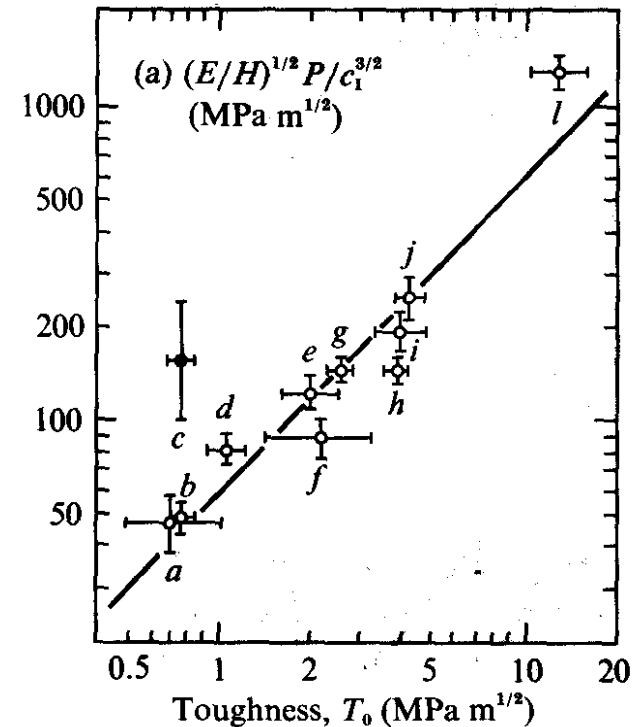
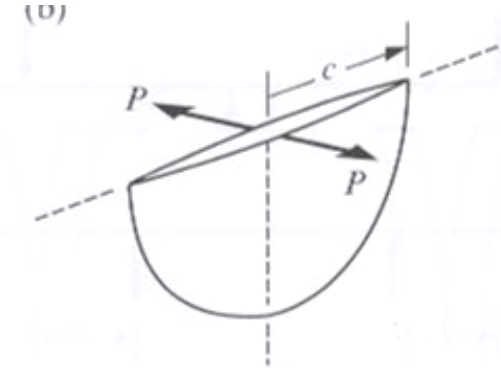
χ Non dimensional parameter depending on Poisson's ratio

- Vickers, cone...

$$K_c = \xi \left(\frac{E}{H} \right)^{1/2} \frac{P}{c^{3/2}}$$

$$\xi = \xi_0 (\cot \Phi)^{2/3}$$

ξ_0 Non dimensional parameter depending on indenter's geometry



Vickers indentation

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